

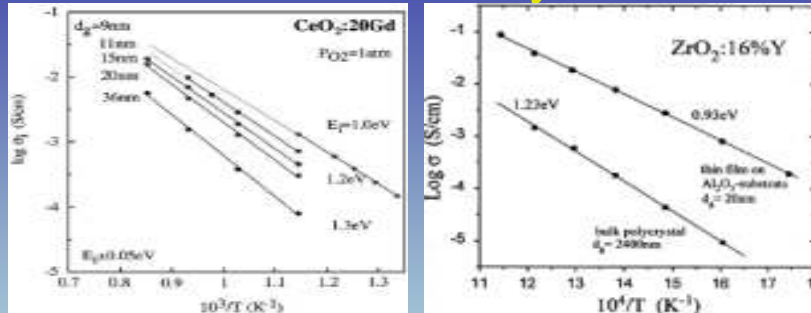
Solid Oxide Fuel Cell Nano- and Micro-Structure Effects

- Nano-phases in SOFCs
 - Motivation: low-temperature SOFCs
 - Nano-electrolyte materials
 - Nano-cathode materials
 - Nano-anode materials

Reduced Temperature SOFCs

- Nano-materials may help enable low-T devices
 - Temperatures 500°C or below possible
 - Nano-materials may be stable at low T
- Why reduced temperature?
 - Lower-cost balance of plant
 - *E.g.* heat exchangers
 - Better materials stability
 - Better, lower-cost materials
 - Seals, interconnects
 - New applications, *e.g.* portable and transportation
- How can nano-materials help?
 - Mitigating reduced reaction/transport rates as T is reduced

Enhanced Ionic Conductivity: Nano-Electrolytes



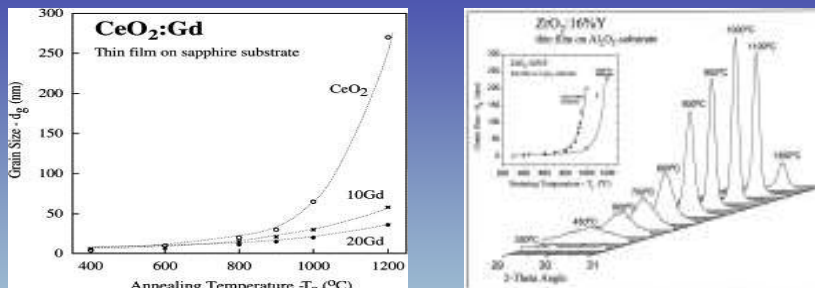
Thin films on sapphire from multiple spin coats from polymer precursor solutions containing nitrates

- Converts to nano-crystalline oxide film as low as 300C in air
- Left: Gd-doped Ceria
- Right: Y-stabilized Zirconia

Suzuki et al., Solid State Ionics, 151 (2002) 111

Kosacki et al., Solid State Ionics, 136-137 (2000) 1225

Nanocrystalline Film Stability



- Average grain size versus annealing temperature (400-nm-thick films)
 - Isothermally annealed for 4 h
 - Grain size from XRD line broadening analysis
- Possible stability at <600°C
 - Long-term (40,000h) stability needs to be established

Issues

- Results not widely reproduced
- Not yet reproduced in bulk materials
- No nano-electrolyte fuel cell results to date
- Long-term stability
- Scientific issues
 - Solute segregation/concentration at grain boundaries
 - “Blocking” species, e.g., Si diluted in nano-crystalline case
 - Space-charge effects/defect gradients
 - Differential transport \perp , \parallel to grain boundaries
 - Altered point defect energetics
 - Enhanced defect mobilities near GBs

Mixed Conductor Electrodes: Nano-Effects

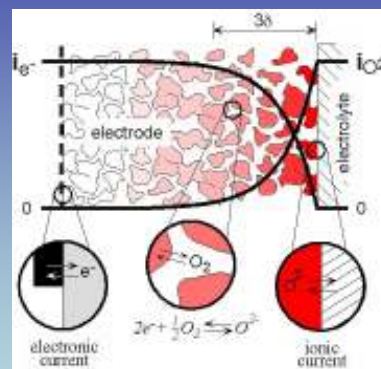
- Model - mixed-conducting cathodes

- Two processes in series
 - Oxygen surface exchange
 - Oxygen bulk diffusion

$$R_{chem} = \frac{RT}{2F^2} \sqrt{\frac{\tau}{(1-\varepsilon)aC_0^2D^*k}}$$

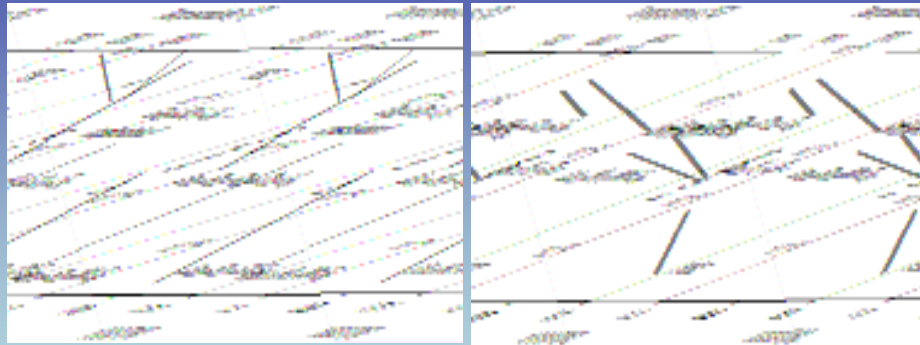
- τ = solid phase tortuosity
- ε = fractional porosity
- a = surface area/volume
- C_0 = oxygen concentration
- D^* = oxygen bulk diffusion coefficient
- k = surface exchange coefficient

- Structure: large a , reasonable τ and ε
- Cathode properties: large D^* , k , along with good electronic conductivity



Adler, Lane, Steele, J Electrochem Soc, 143 3554 (1996).

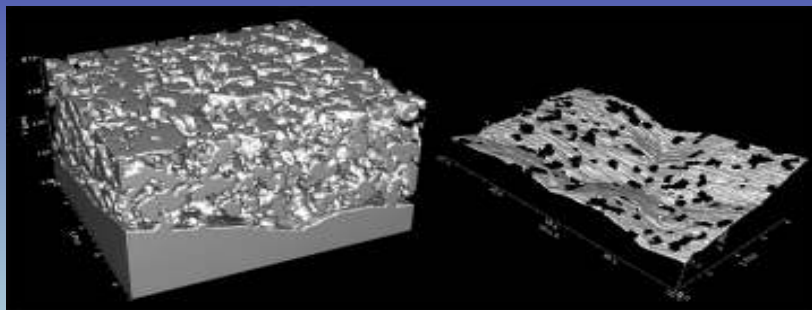
Electrode Transport Data



- Co-based cathodes have highest k values, but D^* not large enough below 700°C
- Cerium has much higher D^*

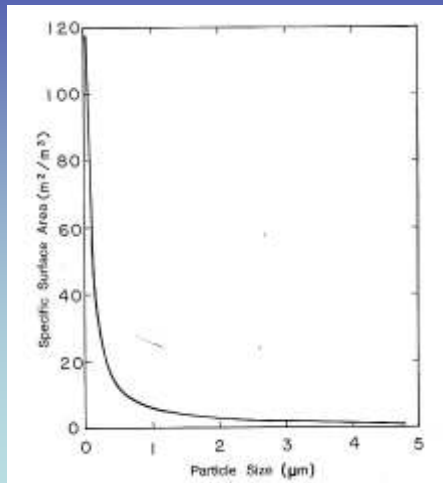
Doshi et al., J. Electrochem. Soc. 146 (1999) 1273.

Example: (La,Sr)CoO₃ Cathode



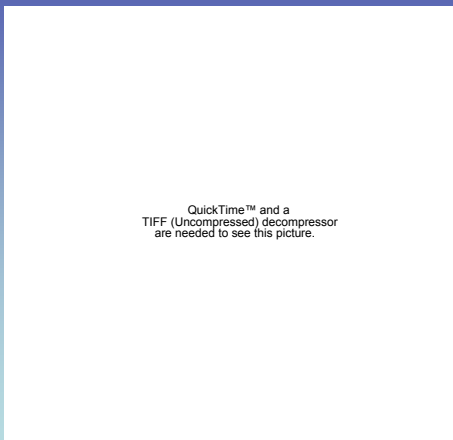
- Feature sizes in the sub-micron range (1100°C)
 - $a = 2.6 \mu\text{m}^{-1}$, $\tau = 1.4$, $\varepsilon = 0.46$
- Calculation yields ASR $\sim 1 \Omega \text{ cm}^2$ at 750°C
 - Agrees with measured ASR's
- Could be improved by appropriate nanostructure
 - a inversely proportional to feature size
 - Needs to be well into nano range, *i.e.*, $\sim 10 \text{ nm}$

Surface Area Versus Particle Size



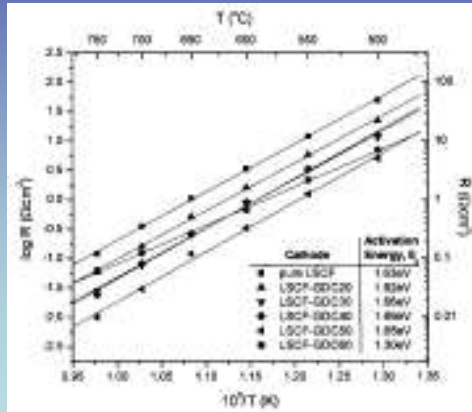
- Surface area per volume a
 - Inversely proportional to average particle radius $\bar{r}_{V/A}$
- $\bar{a}_L \sim 3/\rho \bar{r}_{V/A}$
 - ρ = oxide density
- Very large increases are possible!

Composite “Nano”-Cathodes: LSCF-GDC



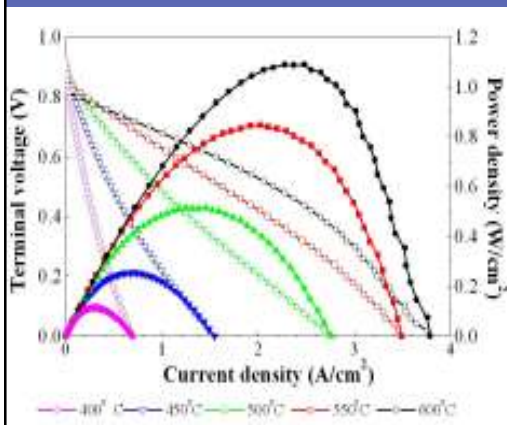
- LSCF = (La,Sr)(Co,Fe)O₃
- Various cross-sectional views of cathode on YSZ electrolyte
- Shows porosity and particle sizes ~200nm
 - Yields high surface-to-volume ratio (a in ALS model)
 - Useful to further decrease size

Polarization Resistance: LSCF versus LSCF-GDC



- Low polarization resistance for LSCF-GDC:
 - 0.3 Ωcm^2 at 600°C
 - 10 times better than LSCF
- Combines desirable properties of two phases:
 - Mixed conductivity of LSCF
 - High oxygen surface exchange coefficient of LSCF
 - High oxygen ion conductivity of GDC
- Feature sizes \sim 200 nm
- Murray and Barnett, Solid State Ionics, 143, 265 (2001)*

Low-Temperature SOFC

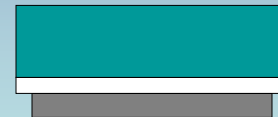


- Ni-GDC/GDC/LSCF-GDC
- Power density:
 - $>1\text{W/cm}^2$ at 600°C
 - 0.5W/cm^2 at 500°C
- Amongst best low T SOFCs
- Employs low-T materials set, but not fully nano!**
- Further reduction to 400°C possible
 - Will require nano-structured materials

Nano-Structured SOFC Anodes

- Goal: improve upon Ni-YSZ anodes
 - Reduced carbon deposition with hydrocarbon fuels
 - Improved reduction-oxidation stability
 - Reduced poisoning from sulfur-containing species
- Alternative materials:
 - Electronically- or mixed-conducting oxides
 - E.g. SrTiO_3 , LaCrO_3 , $\text{Sr}(\text{Mg},\text{Mo})\text{O}_3$
 - Combined oxide-metal
 - E.g., Cu-CeO_2
- However, these materials lack catalyst activity
 - Anode performance worse than Ni-YSZ
 - Possible solution: add nano-phase catalysts

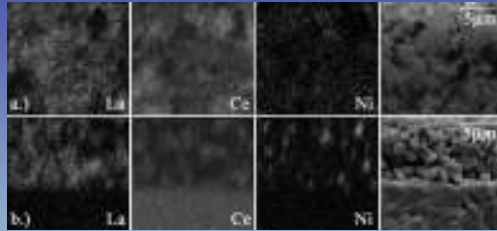
Processing Challenge For Nano-Phase Electrodes



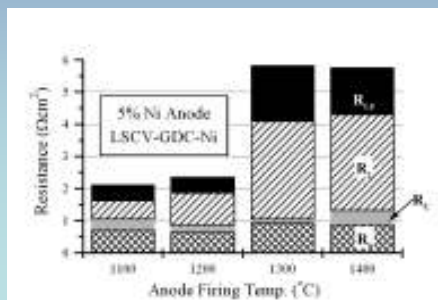
- Cell operation: 500-1000°C
 - Stability a potential issue
- High firing temperatures will coarsen nano-particles
 - Cathode- and electrolyte-supported cells
 - Anode firing temperature:
 - 1000 - 1300°C
 - Anode-supported cells
 - Anode fired with electrolyte at high temperature
 - ~1400°C
- Solution: introduce nano-phases after high-temperature firing steps
 - Wet impregnation (calcination at 600 - 1000°C)
 - Catalyst precipitation

Co-Mixed Anode Example

1100°C



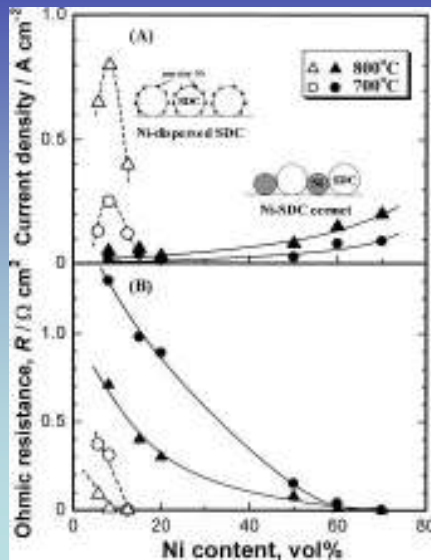
1400°C



- $(\text{La,Sr})(\text{Cr,V})\text{O}_3$ - GDC with nano-NiO
 - Co-fired at high T
- TOP: SEM-EDS
 - 1100°C: Ni particles too small to resolve ($< 1\text{m}$)
 - 1400°C: Coarse Ni particles
- BOTTOM: EIS data versus firing T
 - Increased anode polarization resistance at higher T
 - Cathode and anode components unchanged

Madsen et al., J Power Sources, in press

Ni-Impregnated SDC Anodes

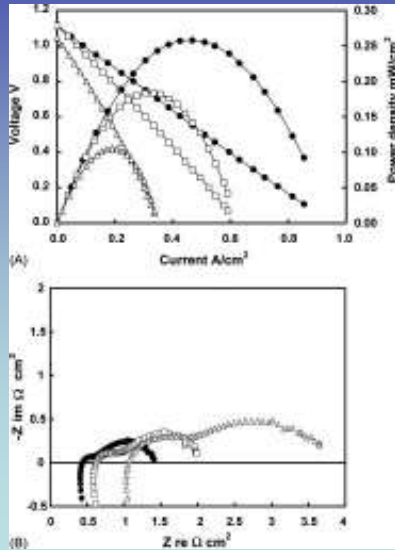


- Improved electrochemical performance (2-3 times higher current density)
- Lesser amount of metals needed than co-mixed anodes
- SEM: feature sizes sub-micron
- Ni activity decreased with time
 - Probably due to sintering

Uchida, Suzuki, Watanabe, Electrochem Solid-State Lett 6, A174 (2003)

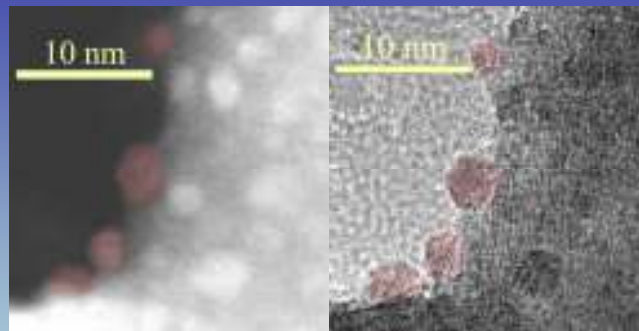
- Similar results for Pt or Ru impregnation into SDC

Performance Degradation During Cell Operation



- YSZ impregnated with 16 vol% Cu and 12 vol% CeO₂
- Current-voltage curves and EIS results (latter at 300 mA/cm²)
 - Measured at 700C
 - Degraded performance with increasing anode firing temperature
 - 700°C for 2 h
 - 800°C for 3 h
 - 900°C for 2 h
- Jung, *J Power Sources* 154, 42 (2006)

Catalyst Precipitation: Ru Nano-Clusters



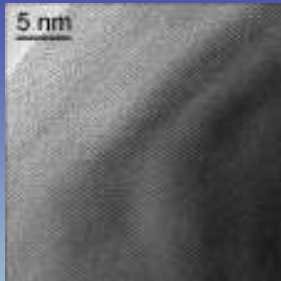
STEM

HREM

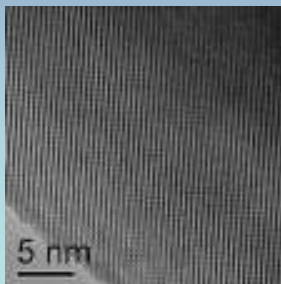
- Images of La_{0.8}Sr_{0.2}Cr_{0.82}Ru_{0.18}O₃ powder after reduction in H₂
 - 45 h at 800°C
- Nano-clusters are ≤ 5 nm diameter
- Cluster lattice fringe spacings matched (110) and (002) hexagonal Ru

Madsen et al., *J Power Sources* 166, 64 (2007)

Control Experiments

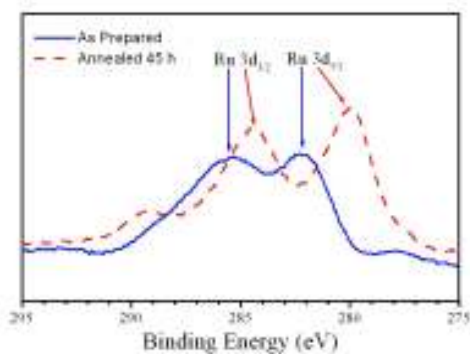


- Unreduced $\text{La}_{0.8}\text{Sr}_{0.2}\text{Cr}_{0.82}\text{Ru}_{0.18}\text{O}_{3-\delta}$
 - No nano-clusters



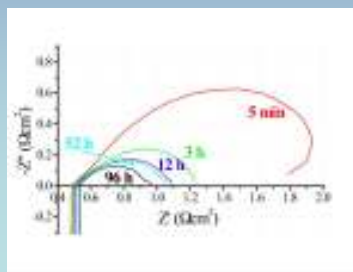
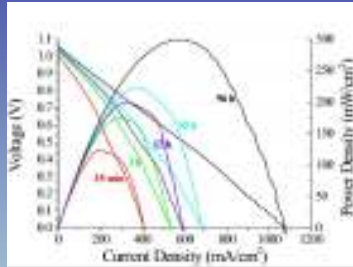
- $\text{La}_{0.8}\text{Sr}_{0.2}\text{CrO}_{3-\delta}$: 45 h at 800°C in H_2
 - No nano-clusters

X-Ray Photoelectron Spectroscopy



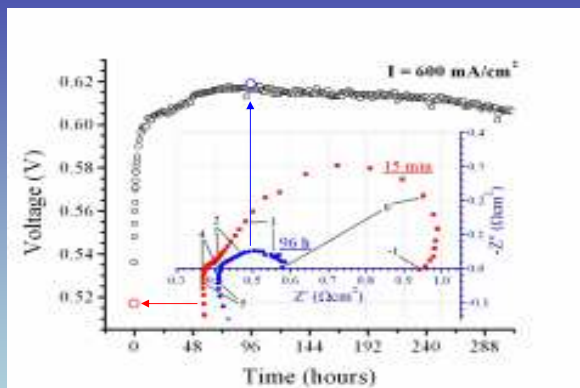
- Comparison:
 - $\text{La}_{0.8}\text{Sr}_{0.2}\text{Cr}_{0.82}\text{Ru}_{0.18}\text{O}_3$ as-calcined in air
 - After 45 h in H_2 at 800°C
- Reduction shifts peaks to lower binding energy
 - Consistent with metallic Ru formation on chromite surface

SOFC Performance: Effect of Anode Ru Doping



- Large increase in power density with Ru-doped anode (750°C)
 - From 0.1 to 0.3 W/cm²
 - From 1.4 to 0.4 Ω cm²
- Only a minor increase with non-Ru-doped anode
 - Much lower power density
- Best anode to date:
 - 0.5 W/cm² at 800°C

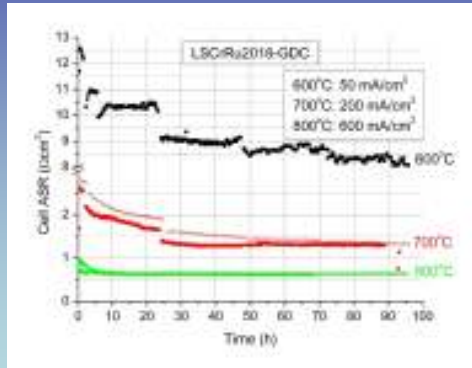
Time Dependence - 800°C



- EIS data:
- LSGM Ohmic loss
 - 0.4 Ω cm²
- Rapid decrease in electrode arcs with time
- Minimum polarization resistance
 - < 0.2 Ω cm²
- Cells limited mainly by LSGM electrolyte

- 0-12 h: Rapid improvement
- 12 to 90 h: Continued improvement
- 90+ h: Slight (2%) degradation
 - Both ohmic and polarization resistances increase

Effect of Precipitation Temperature

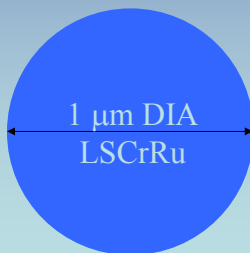


- Kinetics of Ru-induced performance increase
- Wk1 Area-specific resistance decreases with time
 - Slower resistance decrease at lower temperature
 - Consistent with slower Ru out-diffusion from chromite particles
 - Breaks in curves - current stopped for I-V and EIS measurements

Nano-Cluster Kinetics



- Estimate net amount of Ru on surface (45h)
 - Hemispherical Ru ~ 5 nm diameter, $\sim 4 \times 10^{12}$ cm $^{-2}$
 - Corresponds to $\sim 15\%$ of Ru in ~ 1 μ m dia LSCrRu ($y_{Ru} = 0.18$) particle
 - Ru comes from surface layer with thickness $L \sim 25$ nm



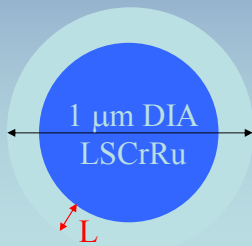
Slide 23

WK1 Data showing rate of increase of performance versus temperature

See slides 18 and 19 for more options

Worawarit Kobsiriphat, 1/18/2007

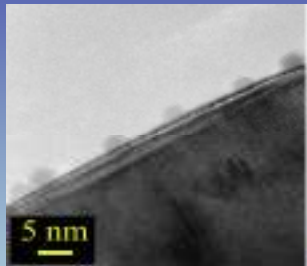
Nano-Cluster Kinetics



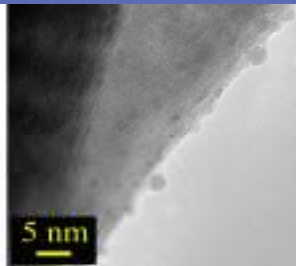
- Estimate net amount of Ru on surface (45h)
 - Hemispherical Ru ~5 nm diameter, $\sim 4 \times 10^{12}$ cm⁻²
 - Corresponds to ~15% of Ru in ~1 μm diam LSCrRu ($\nu_{Ru} = 0.18$) particle
 - Ru from surface layer with thickness $L \sim 25$ nm
- Estimated diffusion coefficient:
 - $D \sim L^2/t \sim 10^{-21}$ m²/s at 800°C
 - Consistent with LaCrO₃ cation diffusion coefficients reported by Sakai, SOFC V, 2001
- Ru out-diffusion may also be limited by increasing B-site deficiency in La_{0.8}Sr_{0.2}Cr_{0.82}Ru_{0.18}O₃

Ru Cluster Size Versus Time

1 h



311 h



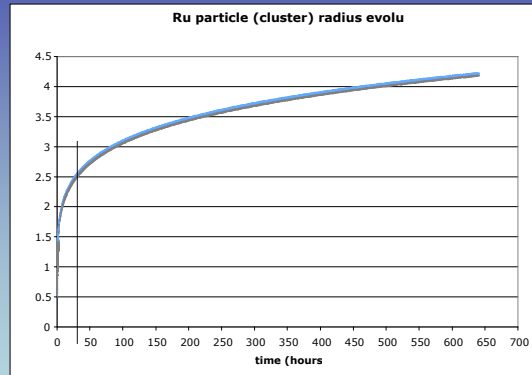
1000 h



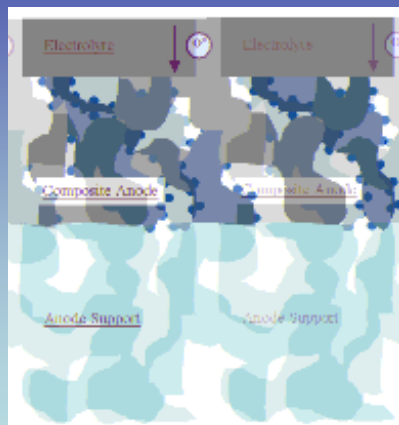
- La_{0.8}Sr_{0.2}Cr_{0.82}Ru_{0.18}O₃ powder reduced at 800°C in H₂
- Measurable coarsening after 1000 h
- May be stable at lower T, e.g. 600 - 700C

Ru Nano-Particle Growth

- Calculated Ru cluster size versus time
 - Assuming simple out-diffusion model
 - No coarsening or cluster growth
 - Diffusion distance estimated as \sqrt{Dt}
- Calibrated using experimental data at 45 hours
 - Particle size of ~ 2.5 nm at 45 h
 - Cluster growth due to continued Ru accumulation
- Phase field calculation under way to consider both coarsening and growth



Comparison with Infiltration



- No additional processing steps required
- Precipitation yields uniformly small (< 5nm) nano-clusters
 - May be smaller than infiltrated materials
- Nano-catalyst selectively added to active layer
 - Infiltration adds material to active layer and current collector / support layer
 - May limit use of expensive materials

Summary and Conclusions

- Reduced temperature operation can widen range of SOFC applications
- Nano-electrolytes have potential for new lower-temperature (<500C) SOFCs
- Nano-cathodes will be critical for achieving good performance at 500C and below
 - Combine nano-structure and composites of optimal materials
- Nano-anodes may provide increased functionality for working with real fuels
 - Hydrocarbons with contaminants